

Hereinafter we will refer to $0.9 \leq x \leq 1.0$ as the first-order region and to $0 \leq x < 0.9$ as the second-order region.

In Fig. 4, the initial pressure derivative of the FM to PM transition temperature, $\partial T_c / \partial P$, is plotted as a function of concentration. The pressure derivatives were determined to within $\pm 0.15^\circ\text{K}/\text{kbar}$. For MnSb our measured pressure derivative of $-3.0^\circ\text{K}/\text{kbar}$ is in good agreement with the value $-3.2^\circ\text{K}/\text{kbar}$ as reported by Hirone *et al.*²⁶ It is observed that $\partial T_c / \partial P$ changes almost precipitously in a very narrow concentration range ($\sim 3\%$) demarcating the first and second-order regions. It should be remarked that the $x = 0.88$ material exhibited no thermal hysteresis at 4.5 kbar -- indicating that the transition remained second-order up to this pressure limit. (According to the Bean-Rodbell model,²⁷ it is possible that a second-order transition can be forced into a first-order transition under sufficient pressure; we shall comment more on this in Sec. IV.)

In Fig. 5 a portion of the temperature versus pressure magnetic phase diagram for MnAs and $\text{MnAs}_{0.9}\text{Sb}_{0.1}$ is shown. Our results for MnAs are in good agreement with the result of Menyuk *et al.*¹ It is observed, as speculated in Sec. I, that the substitution of 10% Sb for As does indeed increase the critical pressure required to stabilize the orthorhombic phase. The increase in critical pressure is approximately 0.75 - 1 kbar.

IV. DISCUSSION

In part A of this section we will discuss the solid solutions which exhibit second-order behavior. The results on these materials will be analyzed in terms of the itinerant FM model as presented in Sec. II. In part B the alloys which exhibit a first-order behavior will be discussed in terms of the model proposed by Goodenough and Kafalas.⁶ In addition, some comments will also be made on the Bean-Rodbell model²⁷ prediction of pressure induced second-order to first-order behavior and on the equivalence of the itinerant electron FM and the Bean-Rodbell models.

A. Second-Order Behavior

In Fig. 6, $\partial T_c / \partial P$ is plotted as a function of T_c for the $\text{MnAs}_x\text{Sb}_{1-x}$ solid solutions in the concentration range $0 \leq x \leq 0.8$. For comparison, the Fe-Ni, Fe-Pd, and Fe-Pt Invar alloys data of Wayne and Bartel²² are included. Similar to the Invar alloys, we observe a T_c^{-1} type of behavior as predicted by Eq. (12) when the second term in Eq. (12) dominates.

The volume derivative of T_c is calculated from $\partial T_c / \partial P$ where the compressibility for the solid solutions was obtained by a linear extrapolation between the values of $2.2 \pm 0.5 \times 10^{-3} \text{ kbar}^{-1}$ for MnSb ²⁸ and $4.55 \times 10^{-3} \text{ kbar}^{-1}$ for MnAs .¹ The values for Γ are given in Table I. We observe that the values of Γ increase with increasing As concentration and that the magnitude of Γ is of the same order of magnitude as the first term in Eq. (10). In previous works on the Invar alloys¹¹ and ZrZn_2 ⁹⁻¹², it was observed that $\Gamma \gg 5/3$ and so the first term of Eq. (10) could be neglected. In the case of the $\text{MnAs}_x\text{Sb}_{1-x}$ solid solutions, this factor of $5/3$ must be included in any calculation of band parameters.

In Table I we give the results of the calculation of \bar{I}_{max} from Eq. (15) for the solid solutions $0 \leq x \leq 0.80$ where we assume $\partial \ln I_b / \partial \ln V = 0$. The quoted error in the compressibility for MnSb will introduce an uncertainty of ± 0.03 in the values for \bar{I}_{max} . We observe that \bar{I}_{max} decreases with increasing As concentration. According to Wohlfarth's²⁹ classification, these values of \bar{I}_{max} indicate that MnSb is approaching a strong itinerant FM, and the solid solutions are becoming weaker itinerant FM's with increasing As concentration. These values of \bar{I}_{max} for the $\text{MnAs}_x\text{Sb}_{1-x}$ solid solutions are comparable with the values for the Invar alloys.³⁰

From Eq. (3) and using the value of \bar{I} and T_c for MnSb from Table I, we calculate $T_F = 1380^\circ\text{K}$. Thus for MnSb we see that $T_c \cong 0.4 T_F$ which indicates the Sommerfeld expansion is converging; however, the convergence is slower than one would desire. For the materials with $x > 0$, the convergence is more rapid than for $x = 0$.